

Magnetism and Microstructure at Relevant Length Scales

Complementary Measurements with Electron and Photon Probes

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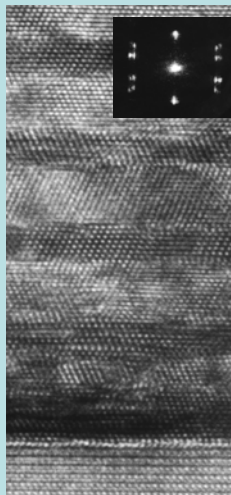
W. Grogger, G. Kusinski and E.C. Nelson - LBNL/UCB
M. E. Gomez and I. Schuller - UCSD
G. Denbeaux and A. Young - ALS/LBNL

Funding: DoE/BES, Max Kade Foundation, IBM, Campbell Endowment at UW

Invited Talk: INTERMAG Meeting, Amsterdam, April 2002

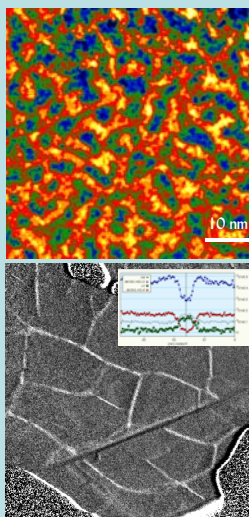
Relevant Microstructural Length Scales & Issues

Structural



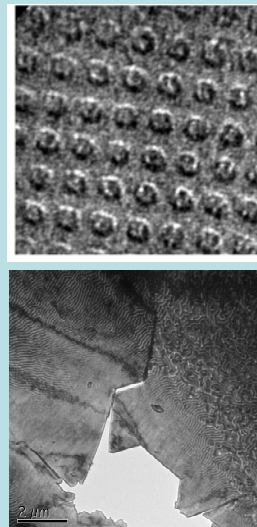
Thin Films & Multilayers
Interface Roughness
Segregation, Mixing etc.

Chemical



Complex Microstructures
Grain Isolation, Second phase
Intergranular coupling etc.

Magnetic



In situ dynamics
Magnetic reversal
Temperature and applied fields


Magnetic Materials

Interactions & Characteristic Length Scales

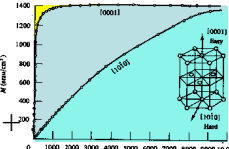
- **Exchange:**

$$E_{ij} = -2 J_{ij} \mathbf{S}_i \cdot \mathbf{S}_j = J_{ij} |\mathbf{S}|^2 \theta^2$$

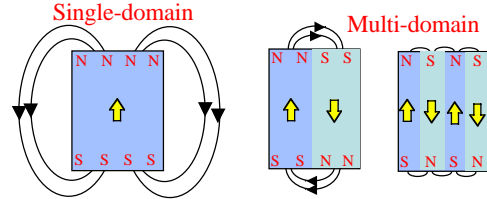
for small θ



- **Magnetocrystalline**

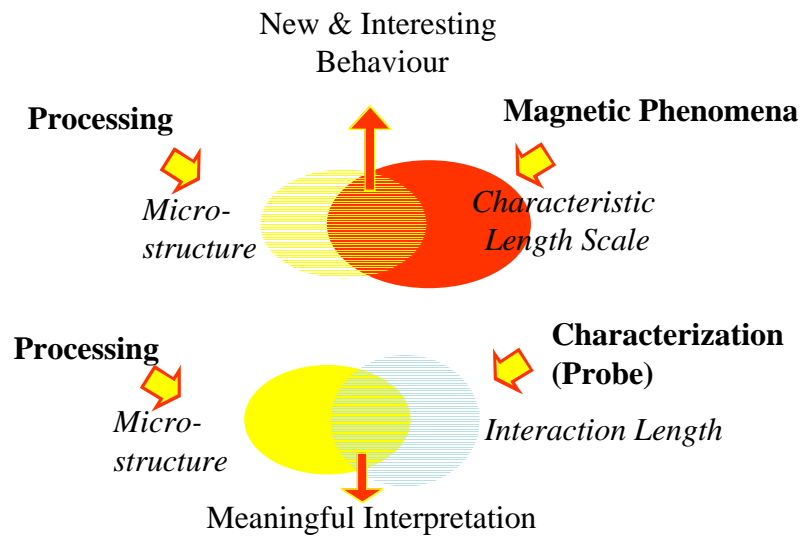
$$E_{mc}^{uniaxial} = K_0 + K_{u1} \sin^2 \theta + K_{u2} \sin^4 \theta + \dots$$


- **Magnetostatic**



Important Length Scales	
Exchange correlation length	$l_{ex} \sim (A / \langle M_s^2 \rangle)^{1/2}$ $l_{ex}(\text{Co}) = 30 \text{ \AA}$
Domain wall width	$\delta = \pi (A/K)^{1/2}$ $\gamma_{wall} = 2 K \delta$
Critical size for single domains	$R_c \sim \frac{\gamma_{wall}}{M_s^2}$
Superparamagnetic limit	$\sim 25 k_B T$ $R_c \sim \left(\frac{k_B T}{K} \right)^{1/3}$

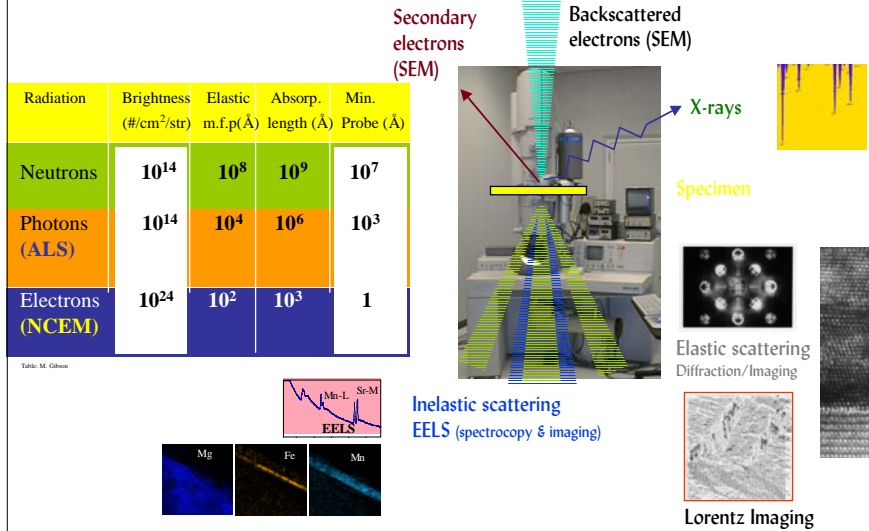
Relevant Length Scales



Review: KrishnanKM in "Magnetic storage systems beyond 2000", NATO/ASI, G. Hadjipanayis ed., Kluwer 2001, pp 251-270

Transmission Electron Microscopy

Beam-specimen Interactions and Signals

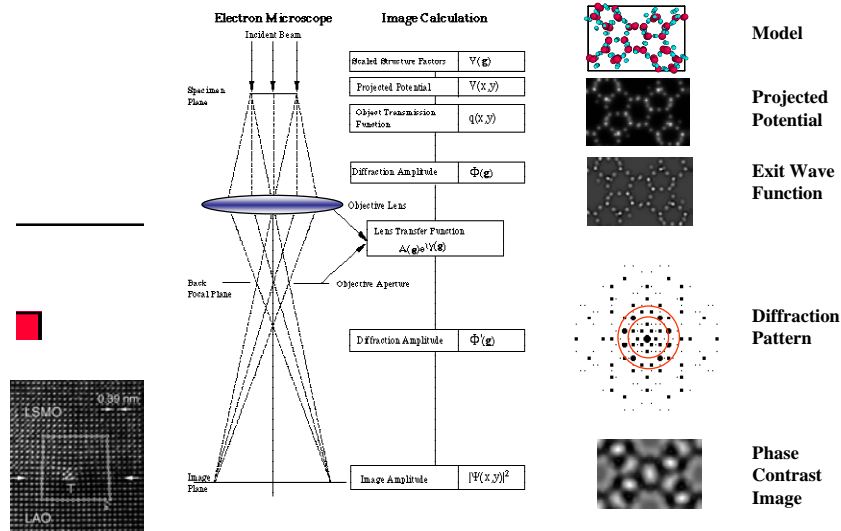


Unsurpassed spatial resolution; Complementary probes - photons and electrons; Buried Interfaces

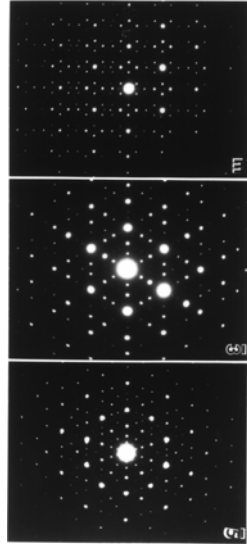
Varela et al. *Phys. Rev. Lett.*, **86**, 5156 (2001); Grogger et al. *Appl. Phys. Lett.*, **80**, 1165 (2002); Lebedev et al. *Phil Mag.*, **A80**, 673 (2000); Kuisinski et al. *Jour. Appl. Phys.*, **87**, 6376 (2000)

High Resolution Electron Microscopy

(Phase Contrast Imaging)

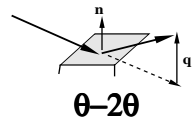


Electron Diffraction in a TEM

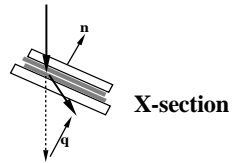
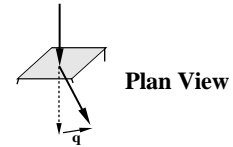


Scattering Geometries:

X-ray

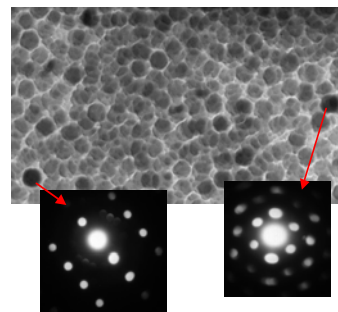
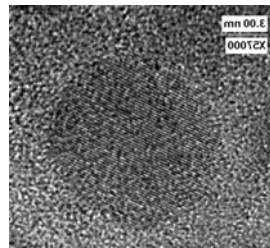
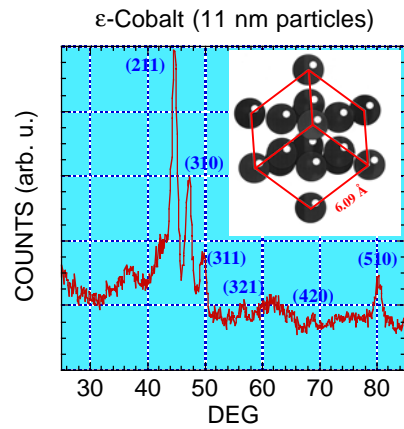


TEM



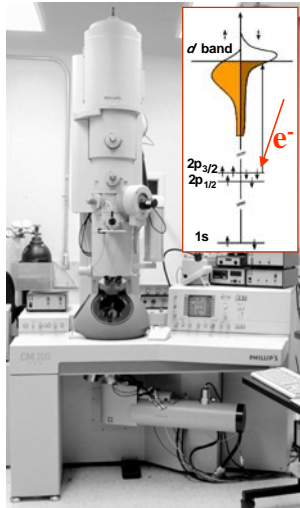
Complementary information for thin films

Structure of Co Nanoparticles



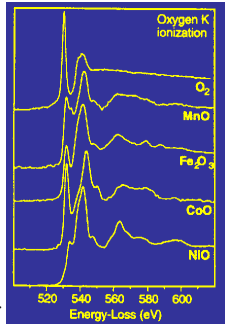
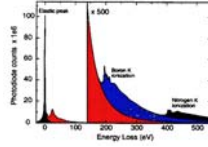
High Resolution EELS Studies in a TEM

Element-specific spectroscopy and imaging at high spatial resolution

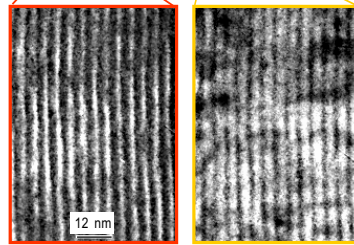
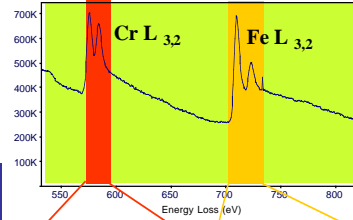


Phillips 200/FEG with Image Filter

Element-resolved local electronic structure

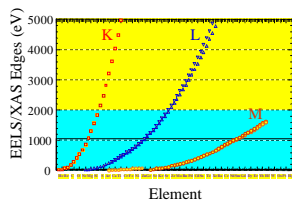


Spatially-resolved measurements



Dipole Selection Rule $\Delta l = \pm 1$ apply

Fine structure in core-loss EELS



Core-level ionization principally determined by a matrix element

$$\langle \Psi_f | \exp(i\mathbf{q} \cdot \mathbf{r}) | \Psi_i \rangle$$

where

Ψ_i is the initial core electron state

Ψ_f is the final ionized electron state

\mathbf{q} is the momentum transfer

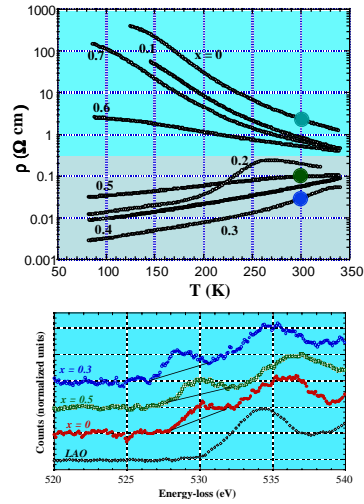
In EELS the scattering is forward peaked, i.e \mathbf{q} is small and the main contribution to the scattering is the linear term

$$\langle \Psi_f | i\mathbf{q} \cdot \mathbf{r} | \Psi_i \rangle$$

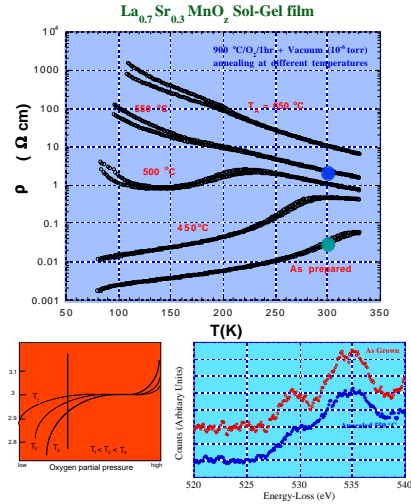
Dipole selection rules $\Delta l = \pm 1$ apply

Electronic Structure of Sol-Gel derived LSMO - EELS Studies

Effect of divalent doping

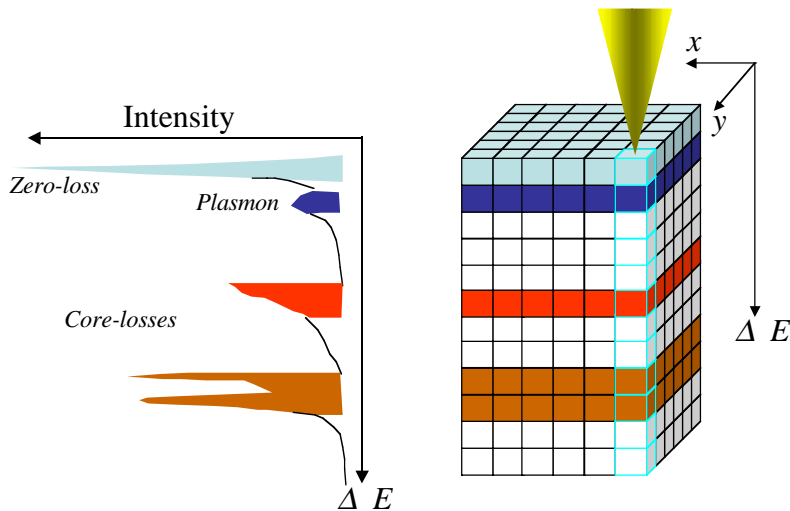


Controlled Oxidation by Annealing

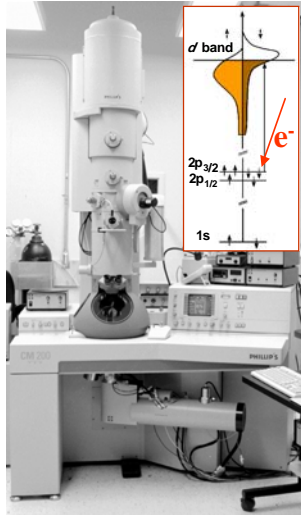


- O_{2p} hole density makes significant contribution to CMR

Energy-loss Spectroscopy and Filtered Imaging

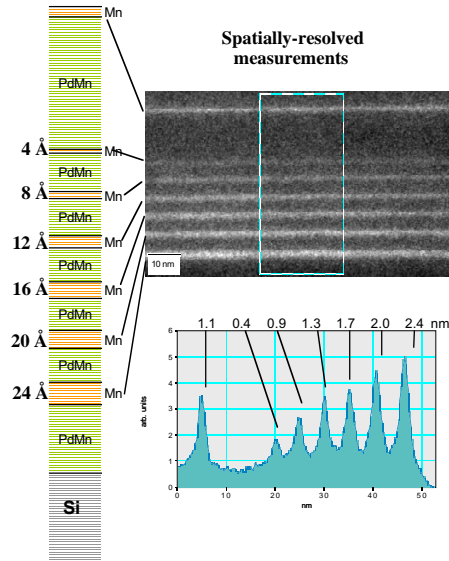


Energy-filtered Imaging in a TEM: Detection Limits

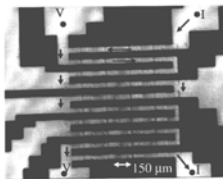
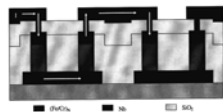


Philips 200/FEG with Image Filter

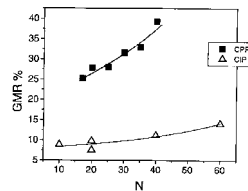
Dipole Selection Rule $\Delta l = \pm 1$ apply



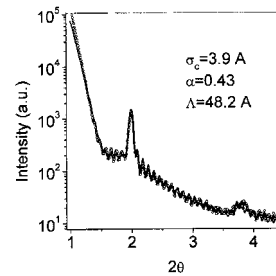
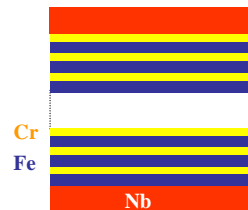
Fe /Cr Multilayers: GMR and Structure



Lithography



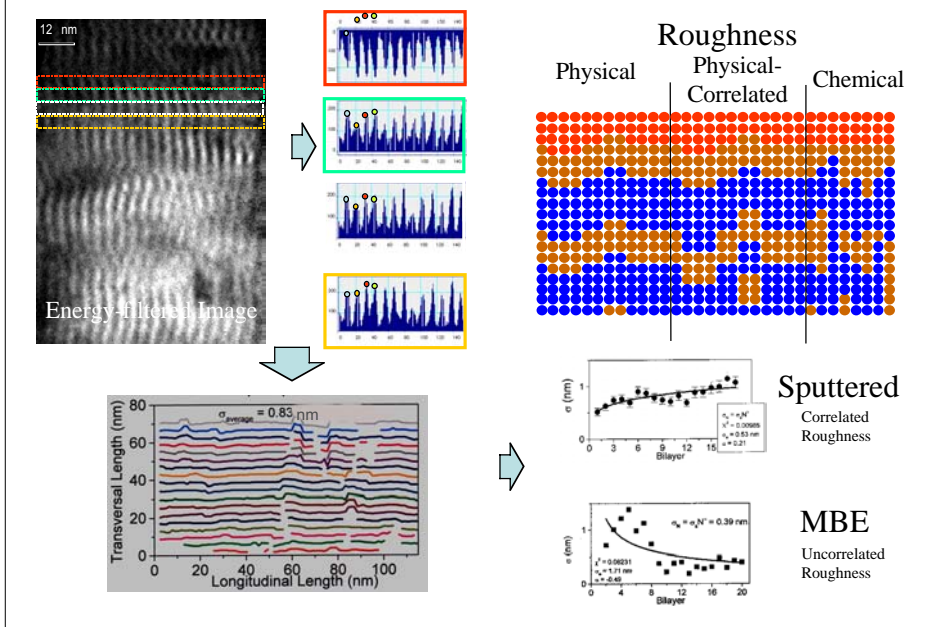
Magnetotransport



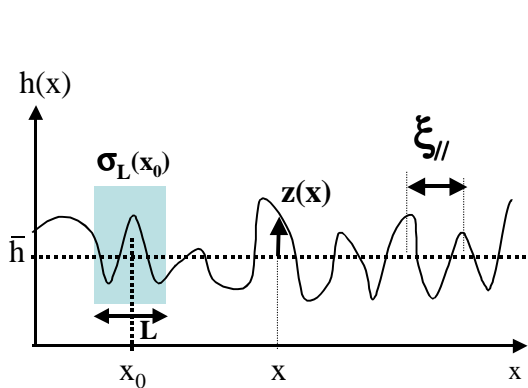
X-ray Scattering

Collaboration: I. Schuller group (UCSD)

Interface Roughness and GMR in Fe/Cr Multilayers



Definitions: Rough interface with various characteristic length scales



Interface Profile $h(x)$
 Average Value $\langle h(x) \rangle$ or \bar{h}
 Height Deviation $z(x) = h(x) - \bar{h}$

Interface width \sim rms roughness
 $\sigma(L) \sim \langle \sigma_L(x_0) \rangle_x$
 where the averaging is done over all points x within L and

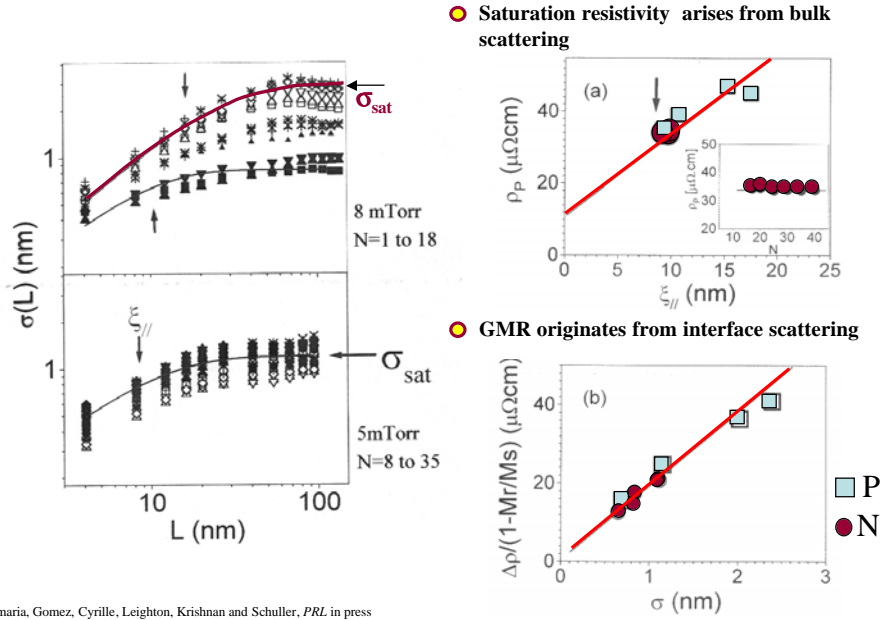
$$\sigma_L(x_0) = \langle |z(x) - z_{av}(L)|^2 \rangle_L^{1/2}$$

$z_{av}(L)$ is $z(x)$ averaged over L

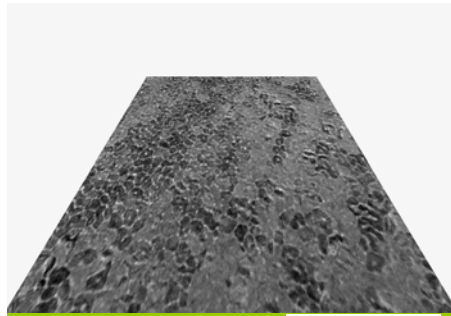
Correlation length, $\xi_{||}$ is extracted by fitting

$$\sigma(L) \sim \sigma_{sat} [1 - \exp(-L/\xi_{||})]$$

Interface dominated GMR in Fe/Cr Multilayers



Quantitative measurements of Cr Segregation



Co-Cr-Pt-X

10% 10%

70% 10%

64% 16%

Magnetic & Recording Characteristics

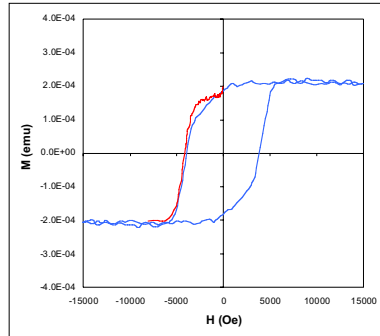
Cr %	16%	10%
H_c (Oe)	4790	4200
SNR(dB)	16.6	12.4
$M_t t$ (memu/cm ²)	0.47	0.69
S^*	0.94	0.93

Quantitative Measurement of Cr Segregation in Magnetic Recording Media Using Energy-Filtering TEM

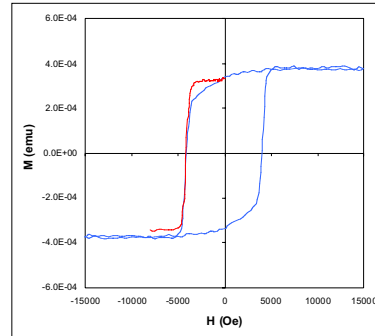
2/3

	SMNR (dB)	dSMNR (dB)	H _{cr} (Oe)	S [*] r	SFDr	Ms (emu/cm ³)
16% Cr	17	+5	4090	0.762	0.26	300
10% Cr	12	0	4120	0.912	0.08	370

16% Cr sample



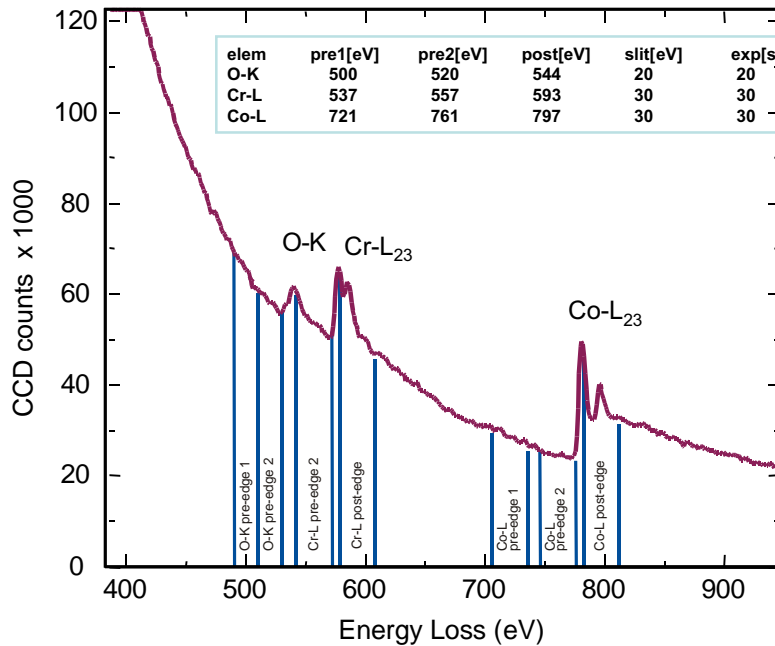
10% Cr sample

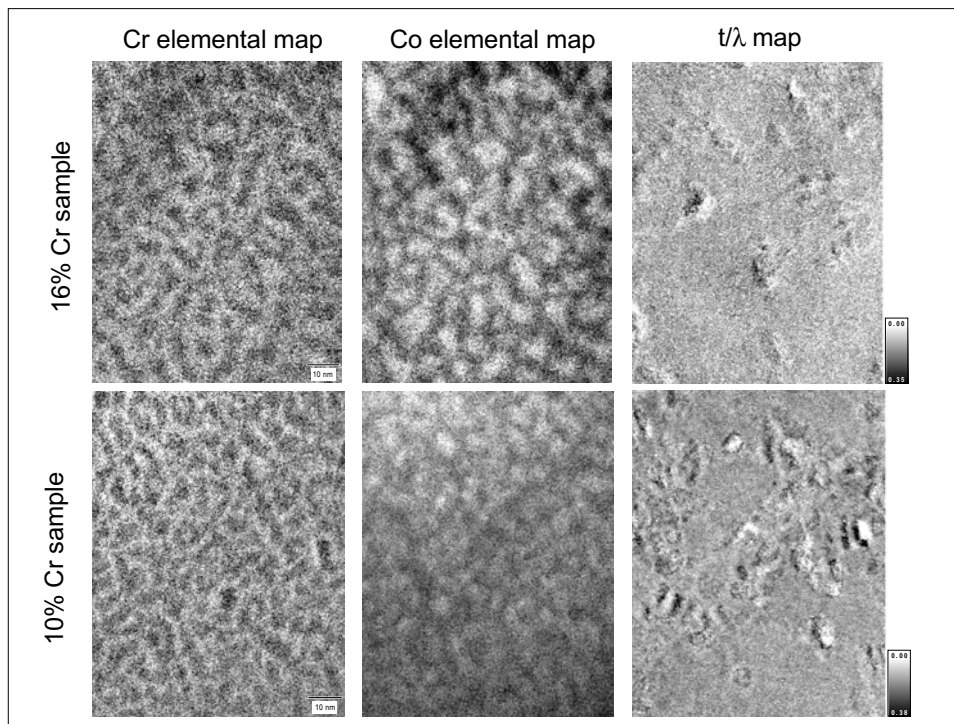


Hysteresis loops (—) and DCD curves (—)

Magnetic data of two CoCrPtX alloys (Co-16Cr-10Pt-10X and Co-10Cr-10Pt-10X). Note the better performance characteristics (e.g. SMNR) for the 16% Cr sample.

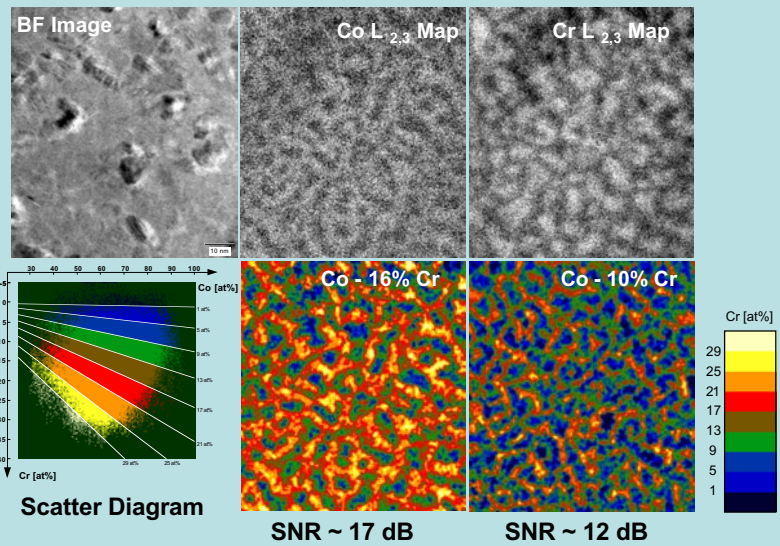
EEL spectrum of CoCr layer





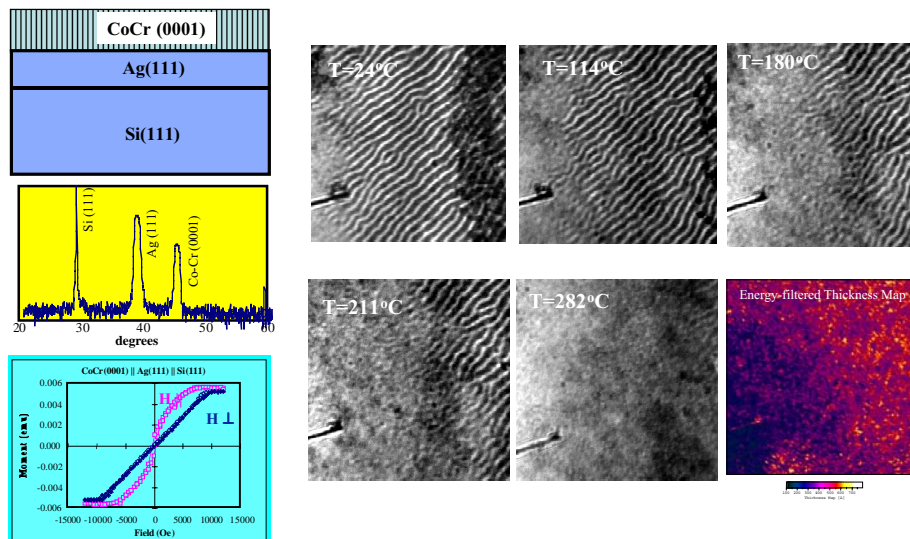
Scatter Diagram Analysis

Quantitative Measurement of Cr Segregation in Magnetic Recording Media Using Energy-Filtering TEM



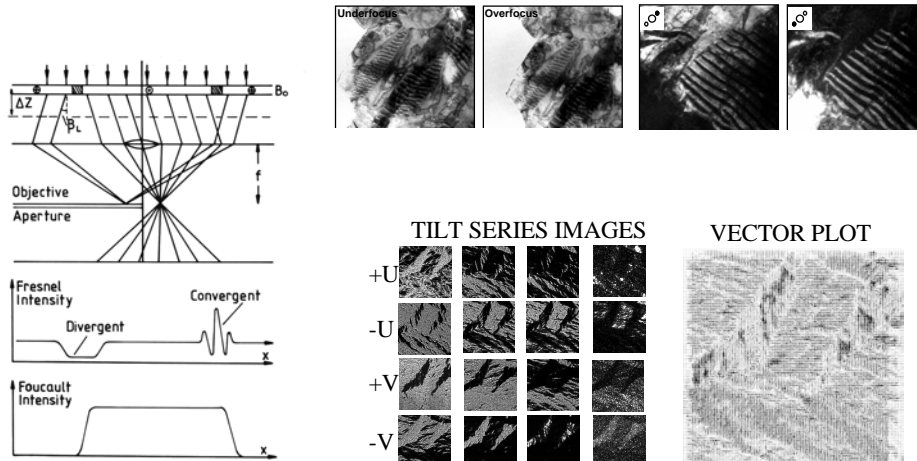
Grogger, Krishnan et al, *Appl. Phys. Lett.*, 80, 1165 (2002)

Epitaxial c-axis oriented Co-Cr films on Si(111): Temperature- & thickness-dependent spin reorientation transitions

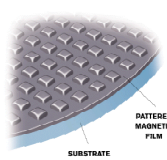


Kanitski, Krishnan & Thomas, *JAP* (in press)

Magnetic Imaging in a TEM



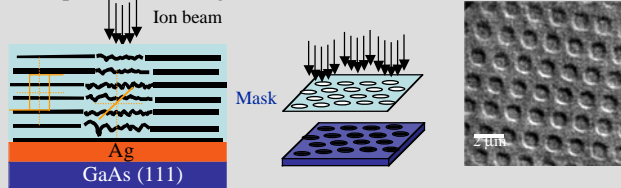
Patterned Media



Features

- Each bit is a single domain
- In theory, stability (volume) scales as $1/N$
- Competitive at 100 Gbits/in², bit size ~ 75-100 nm
- Conventional Lithography Very Expensive

Ion-implantation through stencil masks



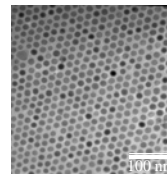
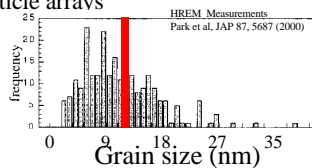
Issues

- Ion-beam solid interaction
- Details of magnetic reversal
- Edge effects

Kusinski et al

Chemical synthesis of nanoscale particle arrays

- Size distribution (narrow)
- Shape (control)
- Self-assembly
- Bulk and Surface structure

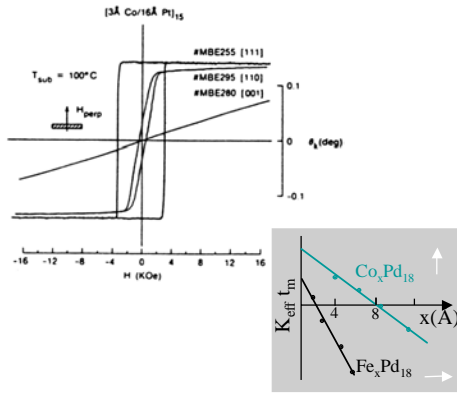
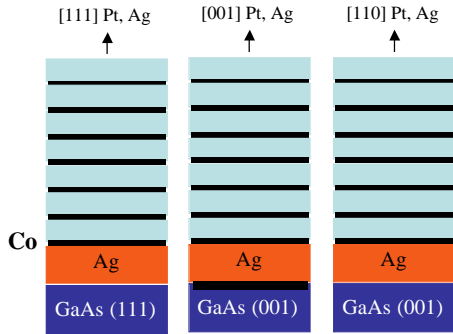


Puntes et al

Surface, Interface and Thin Film Anisotropy

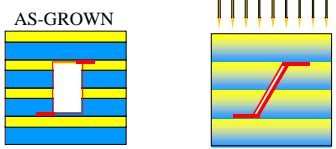
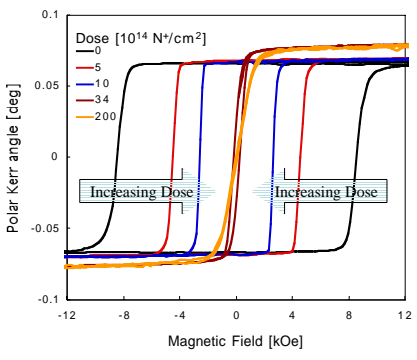
$$E_{an} = K_{eff} \sin^2 \Theta$$

$$K_{eff} = -2\pi M_S^2 + \frac{2K_S}{t_M} + K_V$$



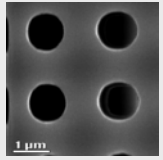
Interface is key to understanding such anisotropy.

Hysteresis loops of the $10 \times (3 \text{ \AA Co} / 10 \text{ \AA Pt})$ multilayers irradiated with an increasing dose of 700keV N^+ ions

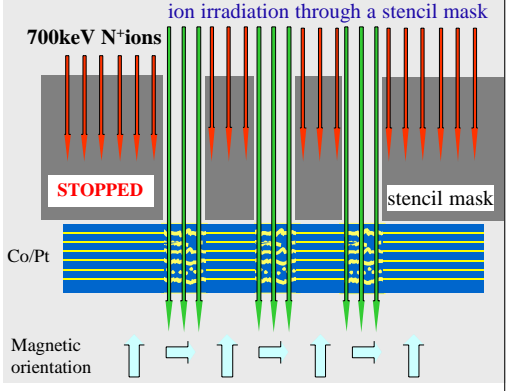


Weller, D. Baglin, J.E.E., Kellock, A.J., Hannibal, K.A., Toney, M.F., Kusinski, G., Lang, S., Folks, L., Best M.E., and Terris, B.D. (2000) Ion Induced Magnetization Reorientation in Co/Pt Multilayers for Patterned Media. *J. Appl. Phys.* 87, 5768-5770.

Ion Beam Patterning of Multilayers



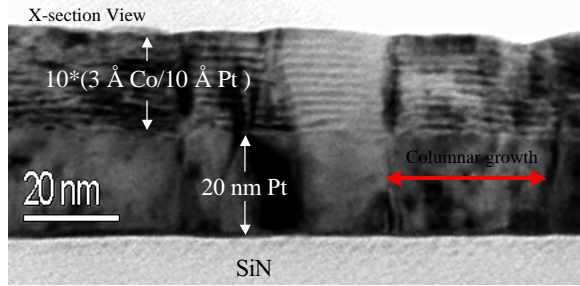
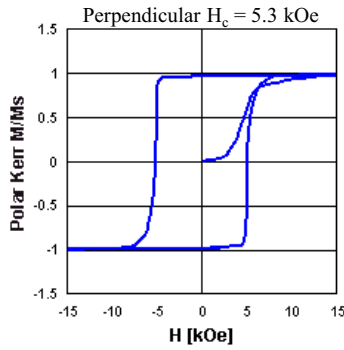
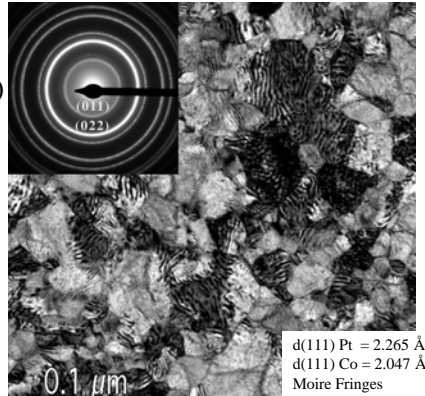
SEM image of the mask with 1 μm holes



Co/Pt Multilayers

- Electron beam evaporated on SiN (windows)
- 200 Å Pt / 10*(3 Å Co/ 10 Å Pt)
- Columnar growth, (111) textured.
- Wide grain size distribution: 50 nm ± 17
- Perpendicular Magnetic Anisotropy

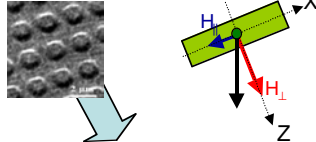
Plan view TEM of ML structure in plan-view



Lorentz (Fresnel) Imaging in a TEM Under Applied Fields

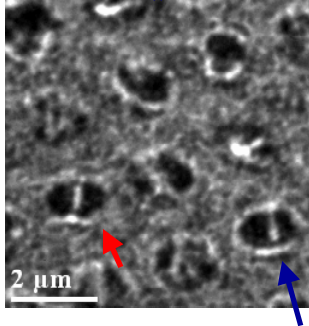
Sample tilt = -15.8°

$H_{\perp} = 1460$ Oe $H_{\parallel} = -413$ Oe



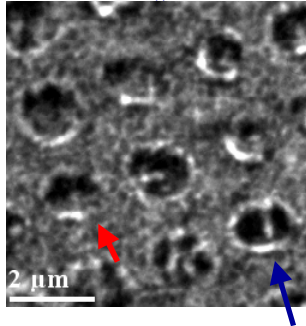
Sample tilt = +2.52°

$H_{\perp} = 1517$ Oe $H_{\parallel} = 67$ Oe



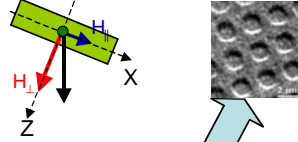
Sample tilt = +3.5°

$H_{\perp} = 1515$ Oe $H_{\parallel} = 93$ Oe



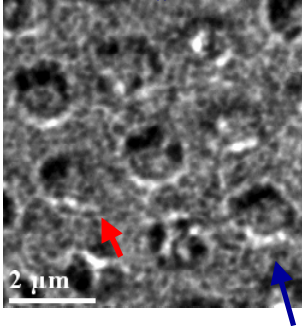
Sample tilt = +15.8°

$H_{\perp} = 1460$ Oe $H_{\parallel} = +413$ Oe



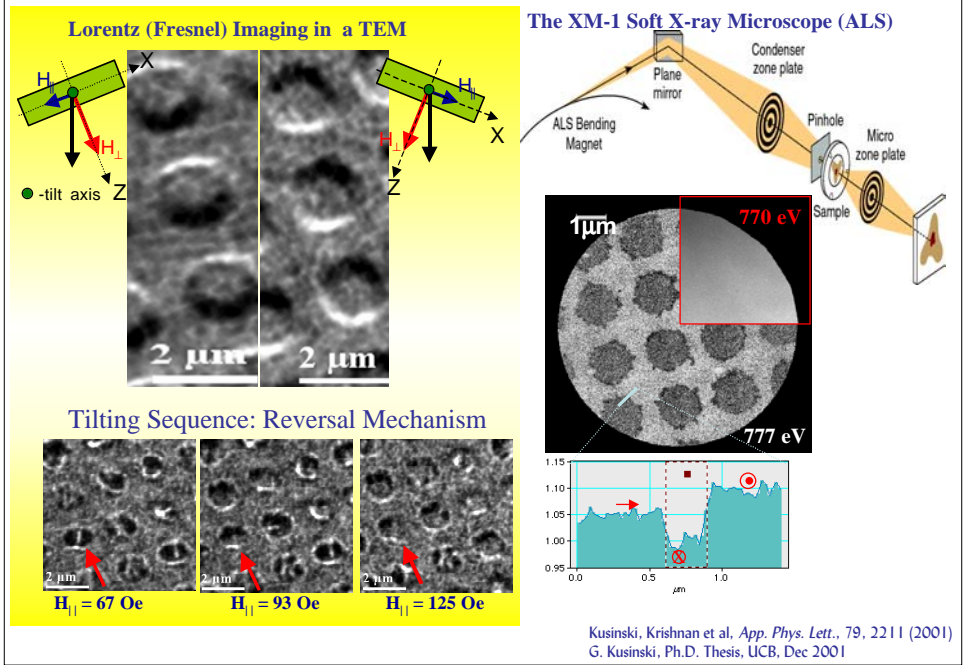
Sample tilt = +4.72°

$H_{\perp} = 1513$ Oe $H_{\parallel} = 125$ Oe



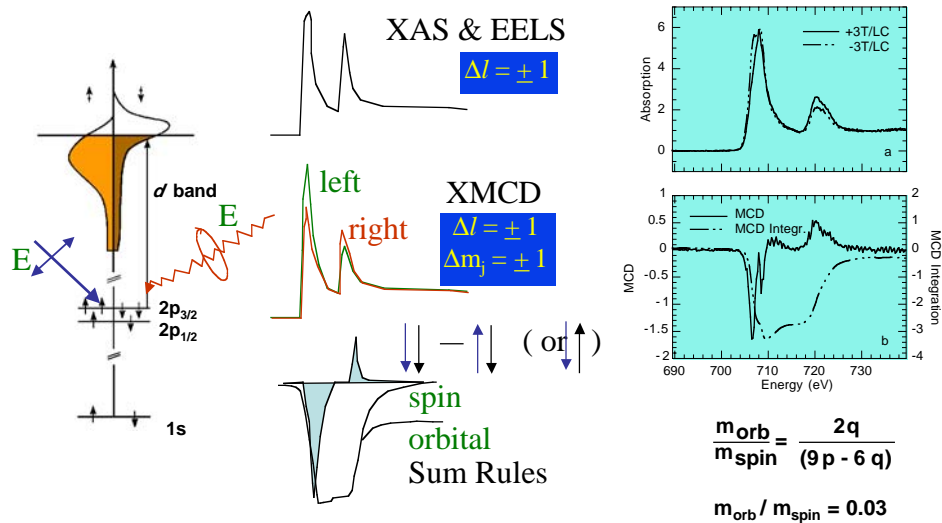
Kusinski, Krishnan et al, *App. Phys. Lett.*, 79, 2211 (2001)

Complementary magnetic imaging of patterned arrays with electrons and photons

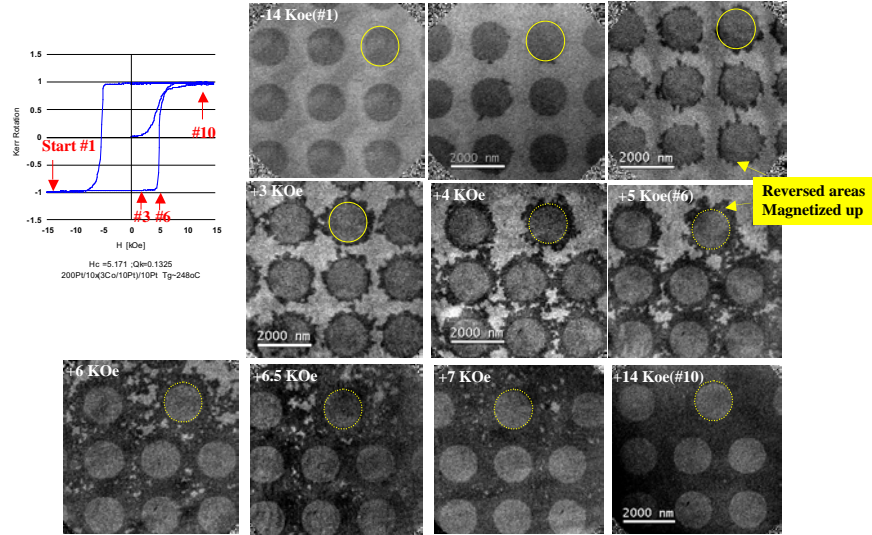


X-ray Magnetic Circular Dichroism

Element Specific, Spin/Orbital Resolved Magnetic Measurements



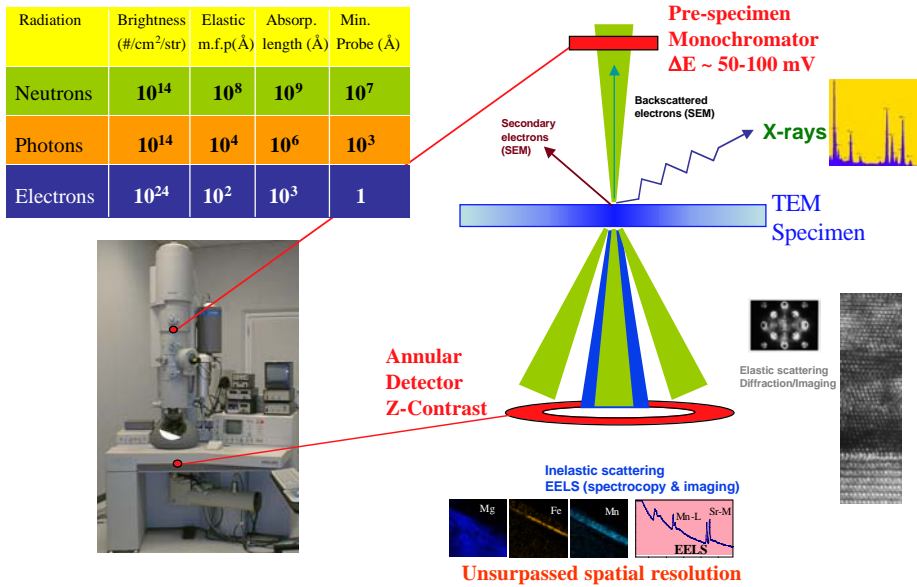
Magnetization Reversal Mechanism: Out of Plane



Kusinski et al, *Appl. Phys. Lett.*, **79**, 2211 (2001)

Next Generation Transmission Electron Microscopes

A Veritable "Synchrotron Beamline" with 1Å resolution in your laboratory



Intermag 2002

Acknowledgements

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Prof. S.-Y. Jeong (7/02-6/03)

Mike Beerman (9/01-)
Yuping Bao (9/01-)
Ji Xiang (9/02-)

Z. Lingley
J. Weaver
H. Sant
J. Kamincek

UCB/LBNL

Dr. W. Grogger
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